#### REPORT DOCUMENTATION PAGE

AFRL-SR-BL-TR-99-

0074

red ( -0188

isting data sources, other aspect of this rts, 1215 Jefferson DC 20503.

Public reporting burden for this collection of information is estimated to average 1 hour per regathering and maintaining the data needed, and completing and reviewing the collection of in collection of information, including suggestions for reducing this burden, to washington Head Davis Highway, Suite 1204, Arlington, VA 22202-4302, and to the Office of Management an 3. REPORT TYPE AND DATES COVERED 1. AGENCY USE ONLY (Leave blank) 2. REPORT DATE 1 Aug 97 to 31 Jul 98 February 1999 Final Technical Report 5. FUNDING NUMBERS 4. TITLE AND SUBTITLE Preparation and Characterization of Superhard Materials of Crystalline Carbon Nitride F49620-95-1-0384 6. AUTHOR(S) Yip-Wah Chung 61102F 2306/BS 8. PERFORMING ORGANIZATION 7. PERFORMING ORGANIZATION NAME(S) AND ADDRESS(ES) REPORT NUMBER Northwestern University Materials and Life Science Bldg 2225 N. Campus Drive Evanston, IL 60208-3108 9. SPONSORING/MONITORING AGENCY NAME(S) AND ADDRESS(ES) 10. SPONSORING/MONITORING **AGENCY REPORT NUMBER** Air Force Office of Sponsored Research/NA 801 N. Randolph Street, Rm 732 F49620-95-1-0384 Arlington, VA 22203-1977 11. SUPPLEMENTARY NOTES 12a, DISTRIBUTION AVAILABILITY STATEMENT 12b. DISTRIBUTION CODE Approved for Public Release; Distribution Unlimited. Α 13. ABSTRACT (Maximum 200 words) Prompted by the prediction of Cohen and coworkers that beta carbon nitride may have properties similar to diamond, there have been many attempts to synthesize this hypothetical metastable material. We successfully synthesized this material via epitaxial stabilization using TiN(111) and ZrN(111) as the growth template. The coatings were CN/TiN or CN/ZrN multilayers grown in a dual-magnetron system. The existence of beta carbon nitride in these multilayers has been confirmed with cross-section electron diffraction, Rutherford backscattering, Raman spectroscopy and near-edge x-ray diffraction. Most important, these multilayers exhibit hardness in the 50 GPa, in spite of the large volume fraction (60-70%) of the softer TiN or ZrN component. Attempts to improve the hardness by using TiB2 (which is harder than TiN and ZrN) as the growth template did not work as well because TiB2 was markedly softened by the presence of nitrogen. Nevertheless, carbon nitride/TiB2 multilayers achieved hardness in the 30-40 GPa range. Lubricated tribo-testing indicated that these very hard coatings exhibit wear coefficients ten times better than conventional TiN coatings. This work was extended to include BNxCy by magnetron sputtering of B4C in a mixed argon/nitrogen ambient. Depending on the process conditions, one can obtain a mixture of h-BN+amorphous carbon nitride to c-BN with excellent electrical and tribological properties. 14. SUBJECT TERMS 15. NUMBER OF PAGES 16. PRICE CODE 19. SECURITY CLASSIFICATION 20. LIMITATION OF 17. SECURITY CLASSIFICATION 18. SECURITY CLASSIFICATION OF ABSTRACT **ABSTRACT** OF REPORT OF THIS PAGE **UNCLASSIFIED UNCLASSIFIED** UNCLASSIFIED III.

F49620-95-1-0384

# PREPARATION AND CHARACTERIZATION

**OF** 

### SUPERHARD MATERIALS OF CRYSTALLINE CARBON NITRIDE

Final Technical Report

May 15, 1995 - November 14, 1998

by

Yip-Wah Chung

Northwestern University

February 1999

Prepared for

AFOSR/NA

Grant F49620-95-1-0384

Dr. Alexander Pechenik

110 Duncan Avenue

Bolling AFB, DC 20332-0001

19990305 004

### 1. Background

This research is prompted by the prediction that a hypothetical material  $\beta$ -C<sub>3</sub>N<sub>4</sub>, which has the same structure as  $\beta$ -Si<sub>3</sub>N<sub>4</sub>, has mechanical properties similar to those of diamond. As research groups around explored different ways of synthesizing this material, there was speculation that  $\beta$ -C<sub>3</sub>N<sub>4</sub> is metastable. While variable non-equilibrium synthesis techniques were used, it was our belief that we must provide some sort of structural template to facilitate its nucleation and subsequent growth.

It turns out that several transition metals (Ti, Zr, Nb and Hf) and their nitrides (TiN, ZrN etc) may be used for this purpose. For example, TiN(111) is hexagonal and is lattice-matched to  $\beta$ - $C_3N_4(0001)$  within a few percent. Other nitrides such as ZrN are even closer. Even if this method of epitaxial stabilization works, there will be a buildup of elastic strain because of the finite mismatch, increasing with the film thickness. Beyond a critical thickness, the system will be unstable. This problem can be solved by depositing alternating layers of these two materials. That is, before the critical thickness is reached, the template layer (e.g., TiN) will be deposited. The whole process can then be repeated until a sufficiently thick film can be synthesized for various applications.

In the course of these investigations, we noted that  $TiB_2(0001)$  is also lattice-matched to  $\beta$ -C<sub>3</sub>N<sub>4</sub>(0001) within a few percent. Bulk  $TiB_2$  is hard (hardness = 35 GPa) and wear-resistant. In fact,  $TiB_2$  is used as a strengthening precipitate for various high performance aluminum alloys. Therefore, we extend our research to include  $TiB_2$  and an important boron-related system, viz. boron nitride. Time limitation precluded us from exploring in detail the boron nitride system. However, some exciting results were obtained that prompted continuing support from industry, as detailed below. In addition, as we continued to expand our research in nanolayered materials, we had a need to extend the capabilities of our existing deposition systems. We were fortunate to be able to acquire a new dual-cathode magnetron sputter-deposition system, funded through the DURIP/AFOSR program. Some of our films were made in this system.

In the following sections, rather than presenting the detailed technical results (which were described in detail in previous reports), we will simply highlight the key conclusion for each material system mentioned above ( $\beta\text{-}C_3N_4$ /TiN,  $\beta\text{-}C_3N_4$ /ZrN,  $\beta\text{-}C_3N_4$ /TiB $_2$ , TiB $_2$ , TiB $_2$ N, and hexagonal BN). A summary of personnel information and publications will be included at the end of this report.

### 2. Experimental Techniques

All coating systems described in this report were synthesized by a dual-cathode magnetron sputter-deposition system. The system operated at a base pressure of mid-10<sup>-7</sup> Torr. To avoid target poisoning, we controlled the flow rate by monitoring the partial pressure of the active gas (nitrogen in this case). Coating properties were optimized by adjusting the substrate bias. We did not deliberately heat the substrate, so the substrate temperature varied according to the process conditions (400-500K typical).

After deposition, coatings were characterized by a wide range of techniques as indicated below:

composition - Auger electron spectroscopy, Rutherford backscattering spectroscopy bonding - FTIR, Raman, near-edge x-ray absorption, electron energy loss spectroscopy hardness/modulus - nanoindentation

wear - pin-on-disk testing

structure - x-ray diffraction, high-resolution transmission electron microscopy surface roughness/morphology - atomic force microscopy

The optimum properties (high hardness, smooth morphology and low wear) were obtained through statistical design of experiments. Results described below represent these optimum materials.

### 3. $\beta$ -C<sub>3</sub>N<sub>4</sub> - related systems

### (a) $\beta$ -C<sub>3</sub>N<sub>4</sub> / TiN

When the carbon nitride thickness is more than 1-2 nm, the hardness of the coating is low (20 GPa or less). Electron microscopy shows that the coating is amorphous. When the carbon nitride thickness is 1 nm or less, the coating is fully crystalline, with hardness in the 50 GPa regime. Electron diffraction reveals extra diffraction peaks matching those from  $\beta$ -C<sub>3</sub>N<sub>4</sub>. In addition, there is a strong correlation between the occurrence of this high hardness and the predominantly (111) texture of TiN. This observation is consistent with our hypothesis that TiN(111) facilitates the nucleation and subsequent growth of  $\beta$ -C<sub>3</sub>N<sub>4</sub>. (b)  $\beta$ -C<sub>3</sub>N<sub>4</sub> / ZrN

The results from  $\beta$ -C<sub>3</sub>N<sub>4</sub> / ZrN superlattice coatings are essentially similar to those of  $\beta$ -C<sub>3</sub>N<sub>4</sub> / TiN. In this case, we made major efforts to determine the structure and local bonding characteristics of the  $\beta$ -C<sub>3</sub>N<sub>4</sub> layers via several techniques. Transmission electron diffraction clearly reveals extra diffraction peaks which can be indexed to  $\beta$ -C<sub>3</sub>N<sub>4</sub>. Raman spectroscopy shows the absence of amorphous features in the 1400-1600 cm<sup>-1</sup> range. Instead, it shows evidence of a C-N stretch around 1100 cm<sup>-1</sup>. Near-edge x-ray absorption shows that the carbon atoms are in the sp<sup>3</sup> state. The latter two observations are consistent with carbon atoms in  $\beta$ -C<sub>3</sub>N<sub>4</sub>. Finally, Rutherford backscattering concludes that the carbon nitride layers have a N/C atomic ratio of 1.3±0.1, consistent with the composition of  $\beta$ -C<sub>3</sub>N<sub>4</sub>. Taken together, these results provide the strongest evidence to date that we have successfully synthesized  $\beta$ -C<sub>3</sub>N<sub>4</sub>.

# (c) $\beta$ -C<sub>3</sub>N<sub>4</sub> / TiB<sub>2</sub>

With the above two systems, the transition metal nitride component is relatively soft (hardness  $\approx$  20-25 GPa), thereby lowering the hardness of the overall coating. By switching to  $TiB_2$ , we were hoping to increase the hardness further. Unfortunately, the two components are not compatible using the dual-cathode magnetron system. To produce carbon nitride, we sputtered a graphite target in an argon/nitrogen ambient. To produce

 $TiB_2$ , we sputtered  $TiB_2$  in an argon ambient. Therefore, in order to deposit β-C<sub>3</sub>N<sub>4</sub> /  $TiB_2$  superlattice coatings, we should have a way to switch the two gases. Such a capability did not exist with our system. Instead, we kept a fixed argon/nitrogen ambient. As discussed in the next section, while oriented, crystalline and hard  $TiB_2$  films can be made by sputtering  $TiB_2$  in argon, presence of nitrogen tends to amorphize and soften  $TiB_2$ . In spite of the various tricks we used to reduce the nitrogen partial pressure in front of the  $TiB_2$  target, we were unable to produce a fully crystalline  $TiB_2$  component in the β-C<sub>3</sub>N<sub>4</sub> / $TiB_2$  superlattice coatings. The hardness of these coatings is  $\approx$  30-40 GPa.

# (d) $TiB_2$ and $TiB_xN_y$

DC magnetron sputtering of  $TiB_2$  by argon can produce crystalline, smooth and highly (0001) textured  $TiB_2$  films, provided that an optimum combination of pressure and substrate bias was employed. Under these conditions, we were able to synthesize  $TiB_2$  films with hardness  $\approx$  45 GPa and r.m.s. surface roughness < 3Å. These films were shown to have wear performance significantly better than TiN. Addition of the smallest amounts of nitrogen to the sputter gas resulted in the reduction of hardness and crystallinity.

# (e) Hexagonal BN<sub>x</sub>C<sub>v</sub>

By performing magnetron sputtering of boron carbide ( $B_{13}C_3$ ) in an argon/nitrogen ambient, it is possible to produce  $BN_xC_y$  thin films. These films have a bluish color, with typical hardness  $\approx 20\text{-}30$  GPa. Transmission electron microscopy studies showed that the films have a turbostratic structure, i.e., hexagonal BN planes lining up perpendicular to the substrate surface. Even without further optimization, these films are extremely useful for dielectric and wear-protection applications. They have electrical resistivity >  $5 \times 10^{10}$  ohm-cm and dielectric strength >  $5 \times 10^6$  V/cm. Further optimization with proper choice of x, y and deposition conditions may yield BN-based coatings with much enhanced properties. The result of this initial work has attracted the interest of IBM, which is interested in developing new materials not only as protective overcoats for hard disk drives, but also for read-write heads. A research contract was awarded by IBM to explore the use

of turbostratic boron nitride and aluminum oxide for specific magnetic recording applications.

#### 4. Personnel Information

Principal investigators:

Yip-Wah Chung, professor of materials science and engineering; Fellow, ASM International; Fellow, Japan Society for Promotion of Science; Board of Directors, American Vacuum Society.

William Sproul, currently with Sputtered Films, Inc, California

Ming-Show Wong, currently chair of materials science and engineering at Hau Tung University, Taiwan.

Students:

Dong Li, graduated in 1995, PhD (MS&E), currently with Motorola

Mei-Ling Wu, expected to graduate in 1999, PhD (MS&E)

Elizabeth Cheang, undergraduate, expected to graduate in 1999, BS (MS&E)

Visiting scholar:

Tong-Jun Zhang, 7/95-5/96, currently at Hauzhong University, China

Postdoctoral fellow:

Ray Chia, 1/96 - 4/96, currently with Western Digital Corporation

#### **Publications**

- 1. D. Li et al, "Structure and hardness studies of CN<sub>x</sub>/TiN nanocomposite coatings", Applied Physics Letters 68, 1211-1213 (1996)
- 2. M. S. Wong, "Crystalline carbon nitride", Physics News (1995), p. 67.
- 3. Y. W. Chung, "Synthesis and surface mechanical properties of amorphous and crystalline carbon nitride coatings", Surface Review and Letters 3, 1597-1602 (1996).

- 4. Y. W. Chung, "Synthesis and properties of crystalline carbon nitride composite superhard coatings", Proceedings of MRS Symposium on Polycrystalline Thin Films II (1996), ed. H. J. Frost, M. H. Parker and C. A. Ross and E. A. Holm, vol. 403, 227-233.
- 5. Mei-Ling Wu, Y. W. Chung, Ming-Show Wong and William D. Sproul), "Preparation and characterization of superhard CN<sub>x</sub>/ZrN multilayers", Journal of Vacuum Science and Technology A15, 946-950 (1997).
- 6. M. L. Wu, W. Qian, Y. Wang, Y. W. Chung, M. S. Wong and W. D. Sproul, "Superhard coatings of CN<sub>x</sub>/ZrN multilayers prepared by DC magnetron sputtering", Thin Solid Films 308/309, 113-117 (1997).
- 7. Mei-Ling Wu, X. W. Lin, Vinayak P. Dravid and Yip-Wah Chung, "Synthesis and tribological properties of carbon nitride and boron-related coatings", Proc. Int. Conf. on Micromechatronics for Information and Precision Equipment, Tokyo (1997), 383-386.
- 8. Mei-Ling Wu, Zunde Yang, Y. W. Chung, Ming-Show Wong and William D. Sproul, "Synthesis of coatings with hardness exceeding 40 GPa by magnetron sputtering", Journal of Tribology 120, 179-183 (1998).
- 9. M. L. Wu, X. W. Lin, Y. W. Chung and V. P. Dravid, "Structural characterization of CN/ZrN superlattice coatings", *Hard Coatings Based on Borides, Carbides and Nitrides: Synthesis, Characterization and Applications*, ed. A. Kumar, Y. W. Chung, and R. Chia, p. 17-24 (1998).
- 10. M. L. Wu, X. W. Lin, V. P. Dravid, Y. W. Chung, M. S. Wong and W. D. Sproul, "Conventional and ionized magnetron sputter-deposition of nanocrystalline titanium diboride thin films", Tribology Letters 5, 131-134 (1998).
- 11. Y. W. Chung and C. Singh Bhatia, "Best bets in protective overcoats for hard disks", Data Storage 5, 47-48 (1998).
- 12. Beizhi Zhou and Y. W. Chung, "Recent advances in the synthesis and properties of amorphous and crystalline carbon nitride", Journal of the Chinese Institute of Engineers 21, 691-700 (1998).